doi: 10.2298/SOS1203365T

UDK 551.4.03;549.514.6;519.224

Effect of Binder Molecular Weight on Morphology of TiO₂ Films Prepared by Tape Casting and Their Photovoltaic Performance

N. Tasić^{1,*}, Z. Branković¹, Z. Marinković-Stanojević¹, G. Branković¹
Department of Materials Science, Institute for Multidisciplinary Research, University of Belgrade, Kneza Višeslava 1, Belgrade, Serbia

Abstract:

Titanium(IV) oxide in the form of anatase has proven to be the best choice for photoanodic material in dye-sensitized solar cells (DSCs). The aim of the work was to study the influence of binder molecular weight on the morphology of deposited films, and consequently, DSC parameters. For this study, five different TiO_2 tape casting slips were prepared from commercially available nanoanatase powder and polyethylene glycol (PEG) as a binder. The process of drying and sintering was carefully designed, so that the organic template was slowly decomposed, leaving favorable crack-free, porous structure. It was found that there is an optimal region of binder molecular weight for obtaining homogeneous, nonagglomerated and porous microstructure which is a necessary condition for application of TiO_2 films in DSCs.

Keywords: Dye-sensitized solar cell, TiO₂, Tape casting, PEG.

1. Introduction

Dye-sensitized solar cells (DSCs) are considered to be a promising alternative to conventional solid-state photovolatic devices, because of their simple assembly and low-cost fabrication [1]. Typical DSC comprises of a photoactive anode, redox electrolyte and counter electrode. Photoactive anode is a semiconducting film deposited on transparent conductive oxide (TCO) glass, covered with the monolayer of dye, for the purpose of visible light harvesting. Upon the introduction of light into the cell, electrons are excited from dye and injected into the conduction band of semiconducting film. The charge carriers are transported to the external load through the anode, while the dye molecules are regenerated via electrolyte, which is present throughout the cell interior. Electrolyte cycle is completed on the counter electrode, usually Pt-coated TCO. Detailed structure and photovoltaic mechanism of DSCs is given elsewhere [2,3].

Special interest is developed towards the synthesis and characterization of an ideal semiconducting material for photoactive anode. It has been concluded that the general demands for highly effective photoanodic films are large surface area, high porosity (50-65%), continuity between its particles, absence of cracks, average thickness 8-12 μ m and good adhesion on TCO substrate [4,5].

Titanium(IV) oxide is widely available semiconducting material, that has been intensively investigated due to its potential in variety of applications [6]. It has emerged as the best solution for photoanodic material in DSCs, because of its good stability under the

^{*)} Corresponding author: nikola.tasic@imsi.rs

irradiation in solution, tunable morphology and physical properties. The highest reported solar-to-current efficiency and long term stability under the irradiation [7,8] are related to

DSCs with TiO₂ as anodic material.

One of the most convenient manners for producing uniform nanostructured layers of desired porosity and thickness is tape casting. It involves the preparation of well-dispersed slips of precursor ceramic powders, with different functional organic additives, including solvents, antiflocculants, binders and plasticizers.

The role of binder in tape casting slips is to provide the polymeric network that holds the entire chemical system together for further processing. Binder polymeric network is the only continuous phase within the slip, which decomposes during the sintering process. The presence of binder strongly affects the rheological properties of the slip as well as strength and density after sintering [9,10].

In this work we have prepared five different TiO_2 slips using polyethylene glycol (PEG) as a binder. PEG is a nonionic surfactant commonly utilized in thin film fabrication as a structure directing agent, owing to its favorable burnout characteristics, low glass transition temperature and good solubility in water and organic solvents [11-15]. We have chosen a wide range of PEG molecular weights (1000-20000) and examined its impact on the porosity, particles size and shape, thickness and adhesion of sintered films, and consequently, the operating parameters of DSCs. Other ingredients of prepared slips include binary solvent system isopropanol/ α -terpineol, stearic acid as antiflocculant and low molecular weight PEG as a plasticizer, and their role in tape casting process is explained elsewhere [9, 16-18].

2. Experimental

2.1. Slip preparation

TiO₂ slips with different compositions were prepared starting from 0.1200 g of titanium(IV) oxide nanopowder (<25nm particle size, spec. surface area 45-55 m²/g, Sigma Aldrich), 1.125 ml of isopropanol (min. 99.7%, AnalaR NORMAPUR, VWR) and 0.375 ml of α -terpineol (96%, Alfa Aesar), which were vigorously grinded together in agate mortar. Then, 0.100 g of stearic acid (Superlab) was added to mixture, and the mixture was additionally grinded for 10 minutes. Finally, after the addition of binder (PEG, molecular weight 1000/4000/6000/8000/20000, Alfa Aesar) and plasticizer (polyethylene glycol, molecular weight 200, Alfa Aesar), slips were deposited on FTO-glass substrates (8 Ω /sq, Xopglass) and microscope glass slides using Elcometer 3580 Casting Knife Applicator. The samples were named based on PEG used for their preparation, for example sample PEG 4000 means the ceramic sample was prepared with PEG 4000 as a binder etc.

As deposited films were dried at 60 °C, and sintered at 475 °C for 1 h, with the temperature raise of 1 °C/min, and a mid-step at 300 °C for 1 h.

Scanning electron microscopy of the films was performed using TESCAN VEGA3 SB microscope, while the X-ray diffractograms were obtained by Rigaku X-Ray Spectrometer KG-3.

2.2. DSCs fabrication and characterization

 3.10^{-4} mol/dm³ films in solution As-sintered were immersed of cis-bis(isothiocyanato) bis(2,2'bipyridyl-4,4'-dicarboxylato) ruthenium(II) bistetrabutylammonium (Dyesol, N719) in CH₃CN/t-butanol (50:50 vol %) for 10 h. The iodideconsisted of 0.50 (the shuttle solution electrolyte 1-methylbenzimidazole iodide, 0.05 mol/dm³ I₂ and 0.5 mol/dm³ lithium iodide in methoxypropinitrele) was dripped between the FTO/TiO₂/dye working electrode and the Pt-coated FTO-glass counter electrode. The counter electrode was prepared by deposition of Pt onto FTO-coated glass (8 Ω /sq, Xopglass,) via thermal decomposition of H_2PtCl_6 solution (2 mg of Pt in 1 cm³ of ethanol) at 450 °C for 30 min. The dye-covered electrodes and Pt counter electrodes were assembled into a sandwich-type cell and sealed with a Surlyn 1702 hotmelt gasket 50 μ m thick. Finally, the assembled cell was filled with the electrolyte containing the Γ/I_3^- redox couple. For the photovoltaic measurements, the DSCs were illuminated with an AM1.5 solar light simulator (SolarLight XPS400) at 100 mW/cm² and their current density-voltage (j-V) characteristics were registered with a Solartron 1287 Potentiostat/Galvanostat.

3. Results and discussion

In has been concluded elsewhere [19], that the anatase crystal phase has superior photovoltaic behavior compared to other TiO₂ phases. Thus, it was of great importance to preserve the anatase presence in sintered films. X-ray diffractograms presented in Fig. 1. confirmed that the chosen sintering regime suppressed the formation of other crystal phases, showing prominent anatase peaks.

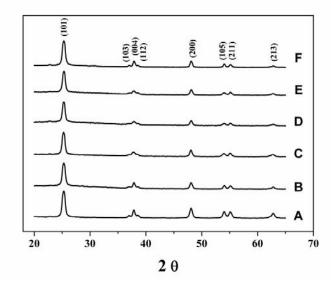


Fig. 1. X-ray diffractograms corresponding to (A) nanoanatase precursor powder, (B)-(F) TiO₂ films prepared from slips with PEG 1000/4000/6000/8000/20000, respectively.

SEM analysis of as-sintered films confirmed the presence of uniform structure in the films with PEG 4000/6000/8000 (Fig. 2). Carefully designed drying and sintering processes allowed the organic ingredients to decompose slowly, leaving porous, crack-free and non-flaking structure. The absence of cracks can also be attributed to chosen powder to binder mass ratio, and to the use of low volatile solvent, α -terpineol [20]. Detailed analysis of micrographs indicates that the average particle size is higher than of the precursor powder. Particles have irregular sphere-like shape, with average size decreasing with the increase of binder molecular weight. The estimated values of the mean grain size are 0.37 μ m, 0.38 μ m and 0.45 μ m, for films with PEG 8000, PEG 6000 and PEG 4000, respectively.

During the process of slip preparation, it was observed that the green sheet with PEG 1000 was too thin, and the final film showed the presence of large agglomerates after sintering. All these observations suggested that the length of chains was insufficient to produce stable viscous polymeric template. This led to the formation of non-uniform films with large agglomerates which were several microns in diameter.

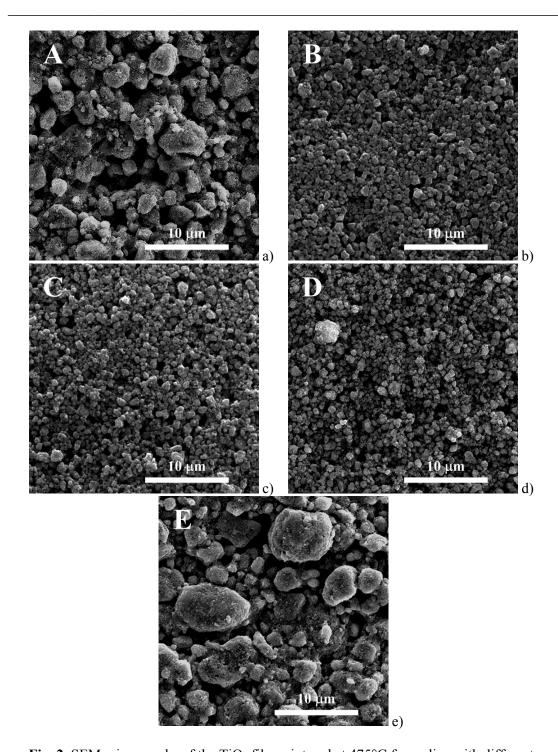


Fig. 2. SEM micrographs of the TiO₂ films sintered at 475°C from slips with different binders: (A) PEG 1000, (B) PEG 4000, (C) PEG 6000, (D) PEG 8000, (E) PEG 20000.

On the other hand, it is considered that a high molecular weight is required to enhance the binder strength [21]. However, long chain binder, PEG 20000, showed poor microstructural properties. Since all the samples contained the same weight percent of binder it could be supposed that because of high molecular weight of PEG 20000 the amount of its molecules and consequently functional groups was too low to homogeneously distribute within the whole slip. Because of that agglomeration of TiO_2 particles occurred. Severe

agglomeration and non-uniform porosity are reasons because the films with PEG 1000 and PEG 20000 were excluded from further investigation of the DSCs.

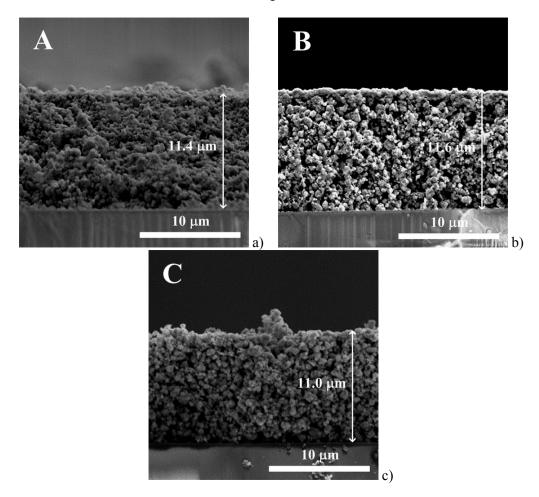


Fig. 3. SEM micrographs of the TiO_2 film's cross-section. Films were deposited on microscope glass slides, with same gap on the tape casting applicator: (A) PEG 4000, (B) PEG 6000, (C) PEG 8000.

Another proof of favorable structure of the films with PEG 4000/6000/8000, comes from SEM micrographs of the films' cross-sections, showed in Fig. 3. Micrographs confirmed the absence of cracks, the presence of pores throughout the TiO_2 films, and interconnected network of TiO_2 particles, which are important characteristics for providing good photovoltaic performances. The film with PEG 6000 showed slightly higher porosity. All three films exhibited satisfactory adhesion, which was tested using a tape stripping method explained elsewhere [22].

Tab. I. Photovoltaic parameters (open-circuit voltage (V_{oc}) , short-circuit current density (j_{sc}) , fill factor (FF) and conversion efficiency (η)) of DSCs with different TiO₂ photoanodes.

TiO ₂ sample	V _{oc} , V	j _{sc} , mA/cm ²	FF	η, %
PEG 4000	0.72	5.65	0.54	2.18
PEG 6000	0.72	7.29	0.50	2.63
PEG 8000	0.71	5.96	0.63	2.64

Photovoltaic performance of DSCs prepared from as-sintered TiO₂ films, acting as photoanodes, was investigated by recording the current density-voltage characteristic (j-V), shown in Fig. 4. Parameters of merit, extracted from curves, are presented in Table I.

As it was expected based on microstructural properties, all three films showed similar efficiencies. Nevertheless, there are some differences in certain parameters. All curves have same values of open-circuit voltage, V_{oc} , i.e. the potential that corresponds to zero current.

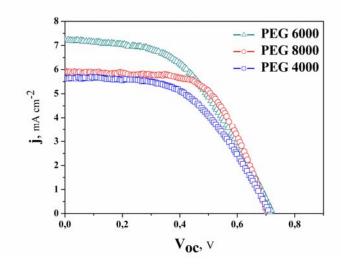


Fig. 4. Current density–voltage characteristic (j-V) of DSCs consisting of TiO₂ photoanodes, prepared from slips containing PEG with different molecular weights.

This is attributed to the same chemical nature of photoanodic material, previously confirmed in XRD analysis. The slight difference between values of j_{sc} , i.e. short circuit density that flows with zero resistance (V = 0), arise from different light trapping within TiO₂ electrodes, but also from active surface area, which are both dependent, between other factors, upon the porosity. Thus, the best result was achieved in samples PEG 6000 and PEG 8000. The sample PEG 6000 showed the highest value of short-circuit current density (j_{sc}), but a fill factor, FF, parameter that describes the "squareness" of the j-V curve, is the highest in sample PEG 8000, due to the presence of clear electron paths, i.e. the lowest internal resistance. Considering all the parameters, extracted from the curves, overall efficiencies were calculated, with the highest value of 2.64% in Sample PEG 8000.

4. Conclusion

Five different TiO_2 slips were prepared starting from commercial nanoanatase powder, and deposited on FTO-glass substrates using tape casting applicator. The correlation between the binder molecular weight and morphology of deposited films, and consequently, the operating parameters of DSCs was established. Films with PEG 4000/6000/8000 showed uniform, crack-free, non-flaking porous nanostructure and good adhesion on FTO substrates. Films with PEG 1000 and PEG 20000 showed structure with large agglomerates, and non-uniform porosity. It was found that there is an optimal region of binder molecular weight for obtaining homogeneous, non-agglomerated and porous microstructures and consequently better solar-to-current efficiency.

The best overall solar-to-current efficiency was achieved with DSSC consisting of TiO_2 photoanode, prepared from slip with PEG 8000 ($\eta=2.64\%$). Sample with PEG 6000 produced higher short current density value ($j_{sc}=7.29$ mA/cm²), due to higher porosity, but

the highest fill factor (FF = 0.63), was recorded for the sample 8000 due to better connectivity between particles.

Acknowledgement

This work was financially supported by the Ministry of Education, Science and Technological Development of the Republic of Serbia (Project number III45007).

We are very grateful to Dr. Márcio de Sousa Góes and Dr. Paulo Roberto Bueno of Institute of Chemistry, Physical Chemistry Department, UNESP, Araraquara, São Paulo, Brazil for measurements of the solar cells.

5. References

- 1. B. O'Regan, M. Grätzel, Nature, 353(1991) 737.
- 2. M. Gratzel, J. Photoch. Photobio. C, 4(2003) 145.
- 3. J. Halme, P. Vahermaa, K. Miettunen, P. Lund, Adv. Mater., 22(2010) E210.
- 4. M. Gratzel, Inorg. Chem., 44(2005) 6841.
- 5. K. Hara, H. Arakawa, Dye-sensitized solar cells, Handbook of Photovoltaic Science and Engineering, Wiley, 2003
- 6. M.J. Šćepanović, M. Grujić-Brojčin, Z.D. Dohčević-Mitrović, Z.V. Popović, Sci. Sintering, 41(2009) 67.
- 7. M.K. Nazeeruddin, F. De Angelis, S. Fantacci, A. Selloni, G. Viscardi, P. Liska, S. Ito, B. Takeru, M. Gratzel, J. Am. Chem. Soc., 127(2005) 16835.
- 8. U. Bach, Y. Tachibana, J-E. Moser, S.A. Haque, J.R. Durant, M. Gratzel, D.R. Klug, J. Am. Chem. Soc., 121(1999) 7445.
- 9. R.E. Mistler, E.R. Twiname, Tape Casting Theory and Practice, Wiley American Ceramic Society, 2000
- 10. S.K. Dhungel, J.G. Park, Renew. Energ., 35(2010) 2776.
- 11. M.A. Khan, M.S. Akhtar, O-B. Yang, Sol. Energy, 84(2010) 2195.
- 12. K. Fan, M. Liu, T. Peng, L. Ma, K. Dai, Renew. Energ., 35(2010) 555.
- 13. B. Guo, Z. Liu, L. Hong, H. Jiang, Surf. Coat. Tech., 198(2005) 24.
- 14. T. Miki, K. Nishizawa, K. Suzuki, K. Kato, J. Mater. Sci., 39(2004) 699.
- 15. K. Kato, K-i. Niihara, Thin Solid Films, 298(1997) 76.
- 16. Z. Jingxian, J. Dongliang, L. Weisensel, P. Greil, J. Eur. Ceram. Soc., 24(2004) 2259.
- 17. Z. Jingxian, J. Dongliang, L. Weisensel, P. Greil, J. Eur. Ceram. Soc., 24(2004) 147.
- 18. K.H. Zuo, D.L. Jiang, J. Zhang, Q.L. Lin, Ceram. Int., 33(2007) 477.
- 19. N.-G. Park, J. van de Lagemaat, A.J. Frank, J. Phys. Chem. B, 104(2000) 8989.
- 20. N. Negishi, K. Takeuchi, Thin Solid Films, 392(2001) 249.
- 21. M.N. Ramahan, Ceramic Processing and Sintering, Marcel Dekker, Inc., 2003
- 22. G. Balasubramanian, D.D. Dionysiou, M.T. Suidan, V. Subramanian, I. Baudin, J.-M. Laine, J. Mater. Sci., 38(2003) 823.

Садржај: Титан(IV)-оксид, у форми анатаса, представља најбољи избор за фотонапонски материјал у фотоелектрохемијским ћелијама побољиваним бојама (енг. dye-sensitized solar cells, DSCs). Циљ рада је био да се испита утицај релативне молекулске масе везива на морфологију депонованих филмова, и последично, на перформансе фотоелектрохемијских ћелија. Припремљено је пет различитих TiO_2 суспензија полазећи од комерцијалног наноанатаса и поли(етиленгликола) (PEG) као

везива, које су затим депоноване на FTO супстрате "tape casting" методом. Процеси сушења и синтеровања су пажљиво дизајнирани, тако да се органска матрица полако разложи, остављајући за собом порозну структуру, без пукотина. Установљено је да постоји оптималан опсег моларних маса везива, за постизање хомогених, неагломерисаних и порозних микроструктура, што је неопходан услов за примену TiO_2 филмова у изради DSCs.

Кључне речи: Соларне ћелије побољшане бојама, TiO_2 , $Tape \ casting \ memoda, PEG.$